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Research paper

Influence on the Phase Formation and Strength of Porcelain by Partial Substitution of Fly Ash Compositions

Nur Azureen Alwi Kutty[1]*, Mohamad Zaky Noh[1], Mohd Zul Hilmi Mayzan[1], Sani Garba Durumin Iya[1]

[1]Materials Physics Laboratory, Department of Physics and Chemistry, Faculty of Applied Science and Technology, Universiti Tun Hussein Onn Malaysia, Pagoh Campus, 84600 Pagoh, Johor, Malaysia
**Corresponding author E-mail: awin_alwi@yahoo.com*

Abstract

This paper presents the study of the influence on the phase formation and strength of the porcelain by the partial substitution of fly ash. The fly ash was calcined at the temperature of 800 °C and partially substituted into feldspar. Each mixture were mixed and pressed into green pellets sintered at different sintering temperature (1100 – 1300 °C) at the interval of 50 °C for 120 min. The compressive strength, crystalline phase and the microstructure of the porcelain were investigated. The optimum physical and mechanical properties were obtained at 5 wt % of fly ash porcelain sintered at 1250 °C. The apparent porosity reaches a minimum value with 0.22 % which is nearly to zero and obtained the highest compressive strength of 105.40 MPa. The XRD results reveal that the highest percentage of mullite was obtained at the substitution of 5 wt % of fly ash with 49.0 %. The glassy phase shows an increasing trend with dissolution of mullite content which affects the strength and microstructure of the porcelain.

Keywords: *Fly Ash; Microstructure; Phases; Porcelain; Strength*

1. Introduction

Porcelain is a traditional ceramic materials which also known as 'whitewares' ceramic due to its white color. Porcelain is systematically produced by three main raw materials of clay, feldspar and quartz with a ratio of 50:25:25 referred as triaxial porcelain. It is comprised of plastic materials, fluxing agent and inert materials as presented in Figure 1 [1]. Although it has been established since 20th century, porcelain is still categorized as higher complexities ceramic in phase development [2]. Generally, quartz (SiO_2) and mullite ($3Al_2O_3{\cdot}2SiO_2$) are the main crystalline phase in porcelain. Iqbal and Lee reported that the equilibrium relation between different phases of the raw material changes due to the vitrification process which required higher temperature thus lead to the development of crystalline phase in porcelain [3]. Literally, the compositional of raw materials of porcelain are widely presented in K_2O-Al_2O_3-SiO_2 phase system [2, 3].

The plasticity characteristic in clay or well known in industry as plastic clay or ball clay helps in giving shape in the forming process [4]. The kaolinite $Al_2Si_2O_5(OH)_4$ (clay mineral) contain of hydroxyl group dehydroxylated at 450~550 °C and transform into metakaolin, $2(Al_2O_3{\cdot}2SiO_2)$ [3]. Above 900 °C, the aluminosilicate spinel ($2Al_2O_3{\cdot}3SiO_2$) phase form by the decompositions of metakaolin. Some of the researcher are reported it as the γ-Al_2O_3 phase [2]. The aluminosilicate phase gives rise to the primary mullite above the temperature of 900 °C which can be seen as the agglomerate of small (< 0.5 µm) mullite crystal [5]. There are two types of mullite which are primary mullite and secondary mullite [6]. The transformation of primary to secondary mullite forms through the reaction of clay relicts with feldspar melts at around 1200 °C and referred as elongated needle-shaped (> 1 µm) mullite crystal which is called as secondary mullite [5]. However, some of the researcher reported that all the primary mullite transformed to secondary mullite at around the sintering temperature of 1350 – 1400 °C due to complete dissolution of quartz grains. This behaviour allows the formation of porcelain microstructure consisting of glass, mullite and quartz [2, 4].

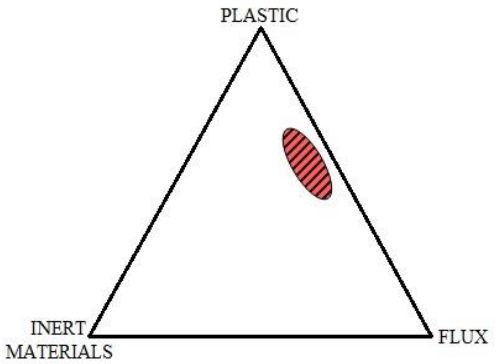


Fig. 1: Ternary diagram of compositional materials in porcelain

Dana and Das reported over the sintering temperature of 1300 °C, the microcracks commonly occurred around quartz grain on the porcelain microstructure due to large thermal expansion coefficient mismatch between quartz and silicate glass matrix [7]. Generally, quartz (filler) is an inert material with high melting point (< 1300 °C) which could reduce the shrinkage and prevent the porcelain bodies to distort during firing process [8]. Meanwhile, feldspar as a flux consists of alkali and alkaline earth metal giving higher vitrification in porcelain. In addition, feldspar minerals also could influence the crystallization behaviour where it could improve the mechanical strength of the porcelain. In emphasizing the strength of the porcelain, several researchers used waste product as part of the raw materials of porcelain in order to improvise its

quality. In fact, the waste materials influenced the crystalline phase behaviour of porcelain. In the present study, feldspar partially replaced by fly ash and sintered in various sintering temperature. The aim of this study is to investigate the phase changes and mechanical behaviour of the porcelain by the substitution of fly ash compositions according to the different sintering temperature.

2. Experimental Procedure

The fly ash was heated through calcination process at 800 °C for 3 h to remove the amount of excess carbon in fly ash and then, mixed with clay, quartz and feldspar for 12 h to obtain the homogeneous mixture for six different porcelain bodies (Table 1). Each mixture was uni-axially pressed into pellet under 3 tonnes pressure using hydraulic press machine (Carver 3851-0) and isostatically pressed afterwards at 100 MPa for higher pressure compaction from all directions using cold isostatic pressing (CP360). All the compacted pellets were placed in the electrical furnace and sintered with a heating rate of 2 °C/min in the temperature range of 1100 – 1300 °C for 120 min. The percentage of shrinkage was measured by measuring the volume of pellets before and after sintering process and the apparent porosity was measured according to ASTM C373-88. The compressive strength was determined using universal testing machine (Testometric) as ASTM C773-88. The chemical analysis of the raw materials was characterized using X-ray Fluorescence (XRF). Characterization on the crystalline phase of the raw materials was carried out using X-ray Diffraction (Bruker D8 Advance) with Cu Kα radiation and analyzed through X'Pert High Score Plus software (PANalytical). Scanning Electron Microscope (SEM) Hitachi-SU1510 was used to determine the microstructure of the samples.

Table 1: Sample compositions mixture (wt. %)

Sample	Clay	Quartz	Feldspar	Fly ash
NP	50	25	25	0
PFA5	50	25	20	5
PFA10	50	25	15	10
PFA15	50	25	10	15
PFA20	50	25	5	20
PFA25	50	25	0	25

3. Results and Discussions

The chemical analysis of the raw materials used is presented in Table 2. Silica, SiO_2 and alumina, Al_2O_3 are identified as the major compound in the raw materials of porcelain. Quartz shows the highest SiO_2 content which is 96.95 %, followed by feldspar and clay with 72.59 % and 56.31 %. Meanwhile, clay and feldspar contain of 39.20 % and 19.28 % Al_2O_3 respectively. The chemical analysis results indicates higher SiO_2 and Al_2O_3 content in fly ash which relevant as the raw materials of porcelain. The presence of 2.25 % of alkali metals oxide (K_2O and Na_2O) and 1.89 % of alkaline earth oxide (CaO and MgO) in fly ash might influence the vitrification process in porcelain bodies.

Figure 2 shows the XRD pattern of the raw materials of porcelain and fly ash. The XRD pattern in Figure 2 (a) indicates clay contain of kaolinite and quartz whereas feldspar contain of albite, quartz, muscovite and microcline. Quartz mineral (SiO_2) is the major crystalline phase in quartz which is in line with the XRF results where SiO_2 was observed as the highest oxide content in quartz. Figure 2 (b) indicates the presence of quartz, albite, mullite and cristobalite in fly ash. Quartz (SiO_2) and albite ($NaAlSi_3O_8$) are observed as the major crystalline phase in fly ash which is 56.6 % and 21.2 % respectively, whereas mullite and cristobalite is 19.2 % and 3 %. As can be observed from Figure 2 (b), the major crystalline phase of fly ash is similar to the feldspar minerals which probably relevant as the flux material. Albite is one of the feldspar mineral so-called as pure sodium Na-feldspar [10]. Tarhan reported that albite is completely dissolved above the sintering temperature of 1100 °C and helps in the formation of glassy phase in porcelain bodies [11].

Table 2: Chemical analysis of the raw materials (mass %)

Oxide content	Clay	Quartz	Feldspar	Fly ash
Na_2O	-	0.07	2.13	0.36
Al_2O_3	39.20	2.53	19.28	26.20
SiO_2	56.31	96.95	72.59	62.90
P_2O_5	0.03	0.01	0.39	0.46
K_2O	1.78	0.20	3.27	1.89
CaO	-	0.04	1.47	1.32
TiO_2	0.96	0.07	0.27	1.66
Fe_2O_3	1.13	0.07	0.35	3.57
MgO	0.31	-	0.07	0.57
L.O.I	0.27	0.05	0.17	1.06

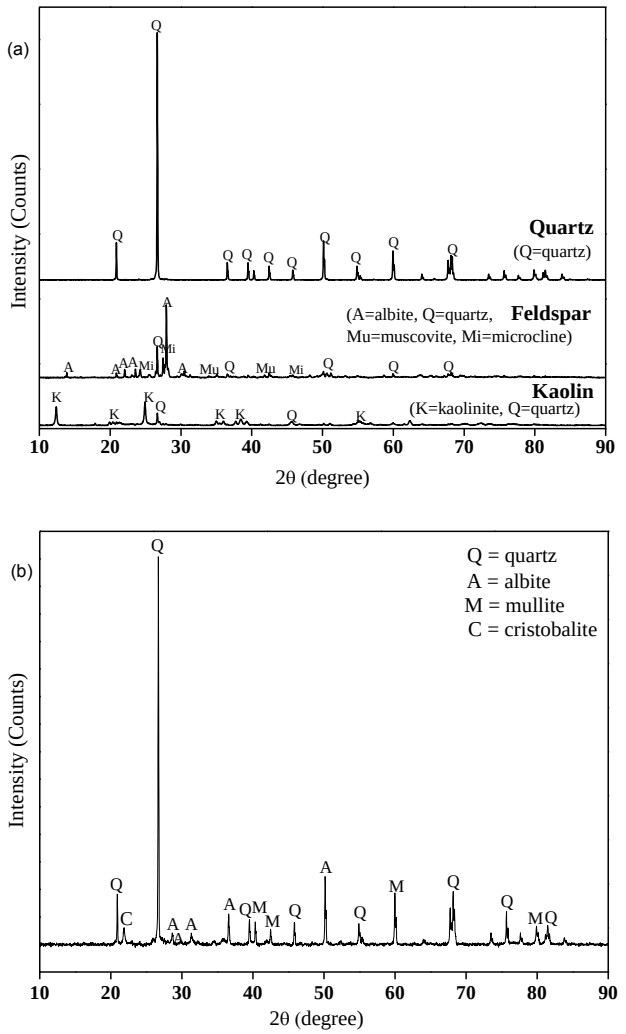


Fig. 2: (a) XRD pattern of the raw materials of porcelain, (b) XRD pattern of fly ash

The variations in the percentage of volume shrinkage are presented in Figure 3. From the figure, it may be observed that the percentage of shrinkage was increased with increasing in fly ash compositions at a sintering temperature of 1100 °C up to 1200 °C. There are no distinct changes on the percentage of shrinkage at lower temperature. However, the shrinkage shows decreasing trend over 1200 °C and abruptly decreased at the sintering temperature of 1300 °C probably due to higher quartz content originated from fly ash. The apparent porosity of the samples in various fly ash compositions and sintering temperature are presented in Figure 4. Higher percentage of apparent porosity presents at lower sintering temperature whereas the apparent porosity shows smaller percentage at higher temperature. It may be observed that

the apparent porosity were decreased at 5 wt% of fly ash and then, increased evenly up to 20 wt % at the entire range of sintering temperature (1100 – 1300 °C). However, over 1200 °C, the apparent porosity were decreased with increasing in fly ash compositions and reached almost zero value (< 0.1%) indicates almost densification achieved at this range of sintering temperature or probably due to over-firing behaviour. Same observation has been made by Das et al. where incorporation of fly ash with porcelain narrowed down the vitrification process [12]. Wang et al. also reported that fly ash porcelain matured earlier compared to the normal porcelain due to the presence of ferruginous and calcareous minerals from fly ash [13].

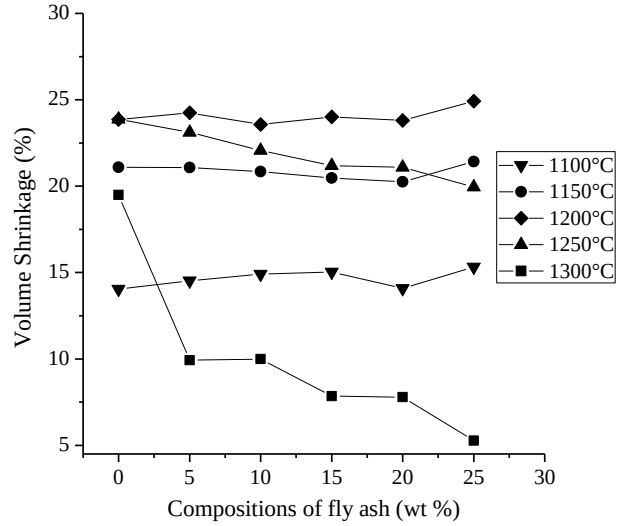


Fig. 3: Variations of the volume shrinkage in various compositions of fly ash and sintering temperature

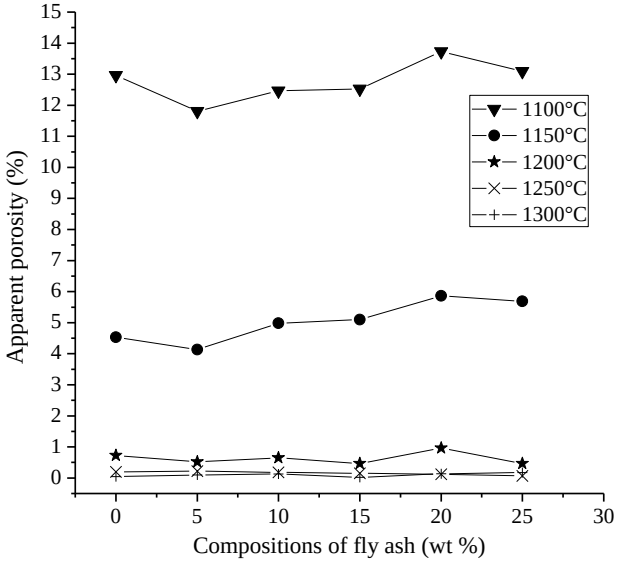


Fig. 4: Variations of the apparent porosity in various compositions of fly ash and sintering temperature

The compressive strength of the porcelain bodies are presented in Figure 5. From the figure, it was observed that the compressive strength was increased at 5 wt % fly ash and then, decreased up to 25 wt % fly ash at the sintering temperature range of 1100 – 1250 °C due to the increasing in glassy phase rather than crystallizations of mullite which may be caused by the presence of alkali content in fly ash [14]. However, the compressive strength was decreased at 1300 °C with increasing in fly ash compositions. It might be related to the stress generated during the cooling process which caused by the larger difference of thermal expansion coefficient between quartz and silicate glassy matrix or probably due to in early stage of overfiring. The maximum strength of the porcelain bodies was achieved at 5 wt % of fly ash with 105.04 MPa sintered at the sintering temperature of 1250 °C which observed as the optimum sintering temperature for the porcelain bodies.

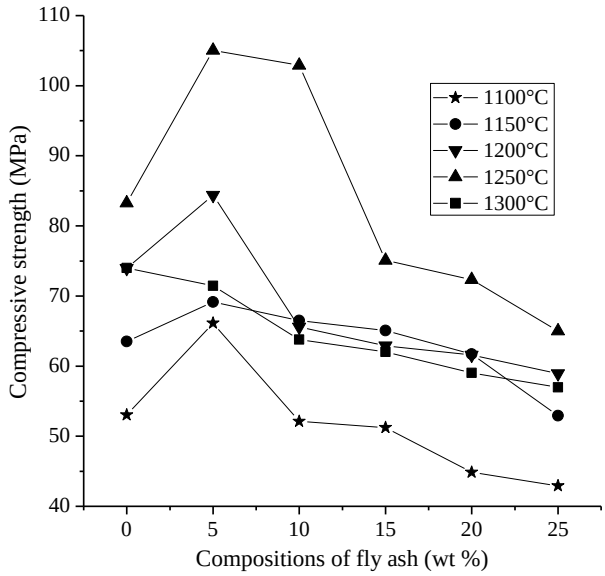


Fig. 5: Variations of the compressive strength of the samples in various compositions of fly ash and sintering temperature

The crystalline phase of the porcelain bodies sintered at 1250 °C was investigated by XRD presented in Figure 6. Meanwhile, the mass percentages of the phase content were quantitatively estimated by XRD presented in Figure 7. Quartz and mullite were identified as the major crystalline phase in normal porcelain and fly ash porcelain bodies. XRD indicates the presence of cordierite in 15 wt % of fly ash porcelain. It was observed that the normal porcelain (0 wt % fly ash) contain of 54 % of quartz and 46 % of mullite. The highest percentage of mullite was observed at 5 wt % of fly ash which is 49 % with decreasing in quartz phase to 51 %. Over 10 wt %, quartz phase shows an increasing trend with decreasing in mullite content. However, 11 % of cordierite $(Mg,Fe)_2Al(Al_3Si5O18)$ or known as iolite was indicates at 15 wt % of fly ash. The expansion of oxygen gas,O_2 or known as "bloating" phenomenon are probably caused by the presence of cordierite where it is contain of Fe_2O_3 and transform into Fe_3O_4 at higher temperature which drastically decreased the densification and strength of the porcelain bodies [9].

The microstructure of 5, 15 and 20 wt % of fly ash are presented in Figure 8. From the figure, the presence of quartz and formation of mullite crystal was observed in Figure 8 (a, b and c). Growth of the cluster needles shaped crystals was observed where the crystals were transformed to the secondary mullite at higher temperature. The extensive micro-crack around the large quartz grains was observed in Figure 8 (a) and (b) due to quartz inversion taking place at the sintering temperature of 573 °C where α-quartz transformed into β-quartz during cooling process and stress generated due to large difference in thermal expansion coefficients between quartz grains ($\alpha \sim 23\times10^{-6}$ °C $^{-1}$) and glassy matrix ($\alpha \sim 3\times10^{-6}$ °C^{-1}) [5]. Excessive glassy phase content and higher formation of closed porosity are also clearly seen in Figure 8 (c) which affects the microstructure in porcelain bodies. Similar observation was made by Luo et al. where dissolutions of mullite increasing the glassy phase content leading to decrease in strength of the samples [15].

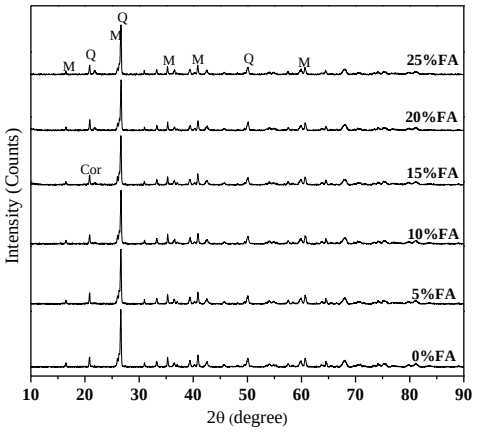


Fig. 6: XRD patterns of fly ash porcelain sintered at 1250 °C (M = mullite, Q = quartz, Cor = cordierite)

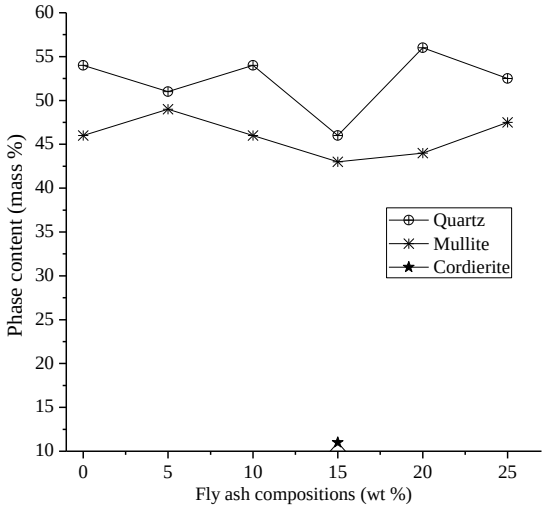
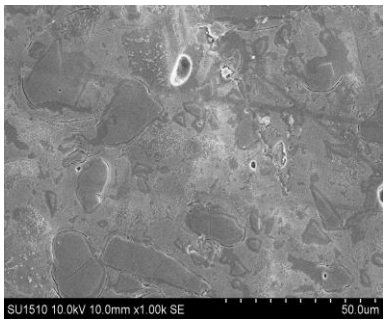
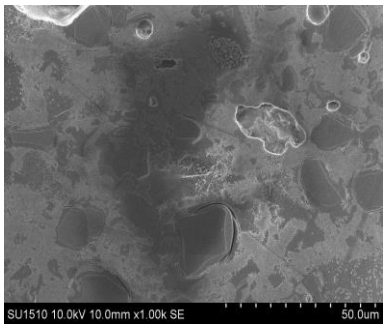


Fig. 7: Variations of phase content of fly ash porcelain bodies sintered at the sintering temperature of 1250 °C

(a)



(b)



(c)

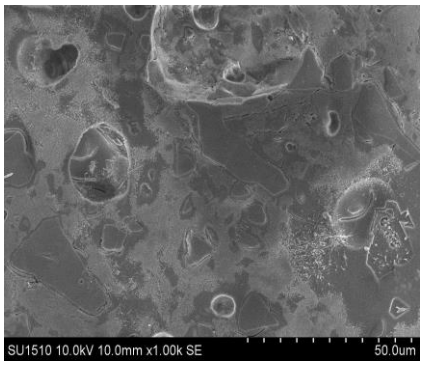


Fig. 8: SEM photomicrograph of fly ash porcelain bodies sintered at 1250 °C; (a) PFA5, (b) PFA15, (c) PFA25

4. Conclusions

Partial substitutions of fly ash on feldspar enhanced the strength and densification of the porcelain. A mixture of 50 wt % of clay, 25 wt % of quartz, 20 wt % of feldspar and 5 wt % of fly ash was identified as the relevant compositions for porcelain sintered at a sintering temperature of 1250 °C. The maximum compressive strength achieved at this composition with 105.04 MPa compared to the normal porcelain with 83.28 MPa. The obtained apparent porosity reach nearly to zero value (< 0.1%) achieving the densification on the microstructure of the porcelain bodies. Mullite and quartz observed as the major crystalline phase in normal porcelain and fly ash porcelain. However, intense interlocking of fine mullite needles in glassy phase matrix observed at 5 wt % of fly ash contributed in achieving higher strength in porcelain. Beyond this composition, the glassy phase increased with dissolution of mullite affects the microstructure, thus decreased the densification of porcelain.

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